

Al-Ti (Aluminum-Titanium)

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Recently, [1990Sch], [2000Ohn], [2001Bra], and [2001Ste] reported new results on the phase equilibria of this binary system. Crystal structure data on the binary phases are summarized in Table 1 [2001Bra]. The complete phase diagram is shown in Fig. 1.

On the Al-rich side, $TiAl_3$ has two crystal modifications: $TiAl_3$ (HT) ($D0_{22}$ -type tetragonal) forms peritectically at 1387 °C [1990Sch] and decomposes eutectoidally at 735 °C [2001Bra]. $TiAl_3$ (LT) forms at ~950 °C and is stable at low temperatures. A metastable form of $TiAl_3$ (AuCu₃-type cubic) also has been reported [2001Bra]. Ti_5Al_{11} is a superstructure based on the AuCu-type tetragonal phase, with the subcell parameters of $a = 0.3953$ nm and $c = 0.4104$ nm at 66 at.% Al and $a = 0.3918$ nm and $c = 0.4154$ nm at 71 at.% Al. It forms peritectically at 1416 °C and decomposes eutectoidally at 995 °C to $TiAl_2$ and $TiAl_3$ (HT). $TiAl_2$ (HfGa₂-type tetragonal) forms congruently at 1215 °C from Ti_5Al_{11} and is stable at low temperatures. A metastable form of $TiAl_2$ (ZrGa₂-type, orthorhombic) was found by [2001Bra] in the as-cast alloys. $Ti_{1-x}Al_{1+x}$ (AuCu-type tetragonal) is stable between 1445 and 1170 °C. Ti_3Al_5 is a low-temperature phase forming below 810 °C. $TiAl$, often designated γ , has the $L1_0$, AuCu-type tetragonal structure with $a = 0.4000$ nm and $c = 0.4075$ nm at 50 at.% Al and $a = 0.3984$ nm and $c = 0.4060$ nm at 62 at.% Al [2001Bra].

On the Ti-rich side, the updated diagram is quite different from the version in [Massalski2] and shows that (βTi) [body-centered cubic (bcc), often denoted β] and liquid undergo a peritectic reaction to yield (αTi) [hexagonal close-packed (hcp), also denoted α] at a high temperature of ~1490 °C. [2000Ohn] found that (βTi) undergoes the $A2$ (bcc) $\rightarrow B2$ (CsCl-type ordered bcc) transition in the temperature range of ~1425 to 1125 °C. The Ti-rich intermediate phase Ti_3Al , commonly called α_2 , has the $D0_{19}$, Ni_3Sn -type hexagonal structure.

References

- 1990Sch:** J.C. Schuster and H. Ipser, Phases and Phase Relations in the Partial System $TiAl_3$ - $TiAl$, *Z. Metallkde.*, Vol 81 (No. 6), 1990, p 389-396
- 2000Ohn:** I. Ohnuma, Y. Fujita, H. Mitsui, K. Ishikawa, R. Kainuma, and K. Ishida, Phase Equilibria in the Ti-Al System, *Acta Mater.*, Vol 48, 2000, p 3113-3123
- 2001Bra:** J. Braun and M. Ellner, Phase Equilibria Investigations on the Aluminum-Rich Part of the Binary System Ti-Al, *Metall. Mater. Trans. A*, Vol 32A, 2001, p 1037-1047
- 2001Ste:** F. Stein, L.C. Zhang, M. Palm, and G. Sauthoff, Al-Ti Alloys with Al-Rich TiAl: Phase Equilibria, Evolution of Phases, and Strength of Lamellar $TiAl + Al_2Ti$ Alloys, *Struct. Intermet. 2001, Proc. Int. Symp. 3rd*, K.J. Hemker, Ed., TMS, 2001, p 495-504

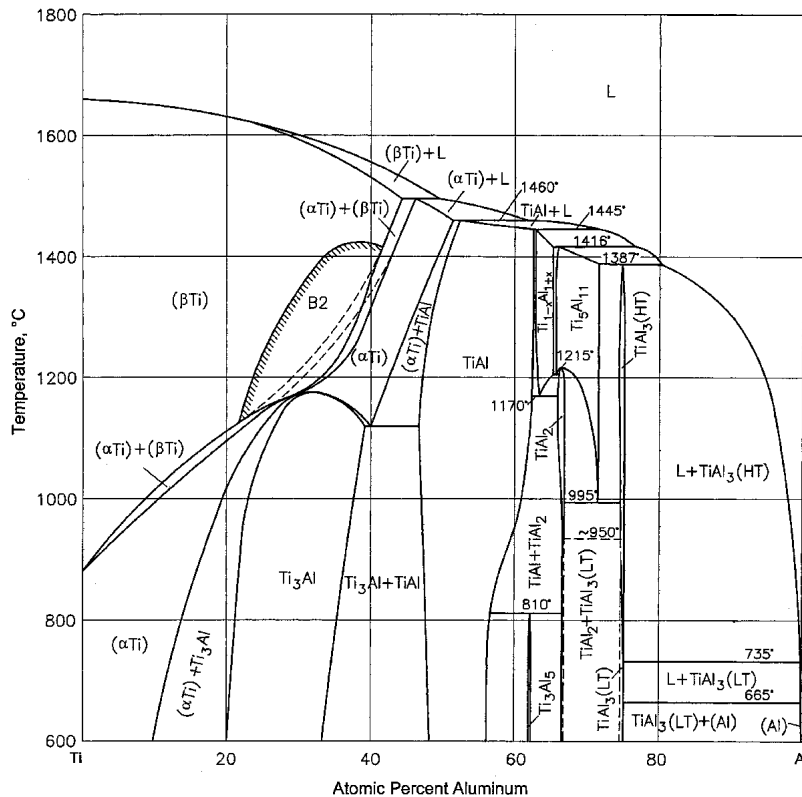


Fig. 1 Al-Ti binary phase diagram [2000Ohn, 2001Bra]

Section II: Phase Diagram Evaluations

Table 1 Al-Ti crystal structure and lattice parameter data

Phase	Composition, at.% Al	Pearson symbol	Space group	Prototype	Lattice parameter, nm
TiAl ₃ (HT)	74.5-75.0	<i>tI8</i>	<i>I4/mmm</i>	TiAl ₃	<i>a</i> = 0.3849 <i>c</i> = 0.8609
TiAl ₃ (LT)	~75	<i>tI32</i>	<i>I4/mmm</i>	...	<i>a</i> = 0.3877 <i>c</i> = 3.3828
Ti ₅ Al ₁₁	66-71	(a)	<i>a</i> = 0.3953 <i>c</i> = 0.4104(b)
TiAl ₂	66-67	<i>tI24</i>	<i>I4₁/amd</i>	HfGa ₂	<i>a</i> = 0.3970 <i>c</i> = 2.4309
Ti _{1-x} Al _{1+x}	63-65	<i>tP4</i>	<i>P4/mmm</i>	AuCu	<i>a</i> = 0.4030 <i>c</i> = 0.3955
Ti ₃ Al ₅	62	<i>tP32</i>	<i>P4/mbm</i>	Ti ₃ Al ₅	<i>a</i> = 1.1293 <i>c</i> = 0.4038
TiAl(γ)	50-62	<i>tP4</i>	<i>P4/mmm</i>	AuCu	<i>a</i> = 0.4000 <i>c</i> = 0.4075(c)
Ti ₃ Al(α_2)	~20-39	<i>hP8</i>	<i>P6₃/mmc</i>	Ni ₃ Sn	<i>a</i> = 0.5782 <i>c</i> = 0.4629

(a) Tetragonal. (b) Subcell parameters at 66 at.% Al. (c) At 50 at.% Al